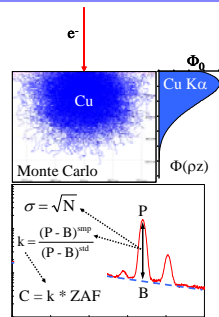


Electron-Probe Microanalysis (EPMA): Summary for Microprobe Lab Demo

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EPMA Summary Resources

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Books on EPMA and Quantitative Analysis

- ◆ Scanning Electron Microscopy and X-ray Microanalysis, 3rd Ed., Goldstein et. al. 2003 Plenum. Mandatory! – Previous editions very good
- ◆ Electron Probe Quantitation, Heinrich and Newbury. 1991 Plenum. "The green book".
- ◆ Electron Microprobe Analysis, 2nd Ed., Reed. 1997 Cambridge. Very good.
- ◆ Electron Microprobe Analysis and Scanning Electron Microscopy in Geology, 2nd Ed. Reed. 2006 Cambridge.
- ◆ X-ray Spectrometry in Electron Beam Instruments, Williams, Goldstein, and Newbury. 1995 Plenum.
- ◆ Quantitative Electron-Probe Microanalysis, Scott and Love, 2nd Ed., 1995 Prentice Hall.

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Websites with EPMA Information

- ◆ Paul Carpenter's lab web site
<http://xraysrv.wustl.edu/>
- ◆ John Fournelle's lab web site (see course materials)
<http://www.geology.wisc.edu/~johnf/sx51.html>
- ◆ John Wittke's lab web site
<http://www4.nau.edu/microanalysis/Microprobe/Probe.html>

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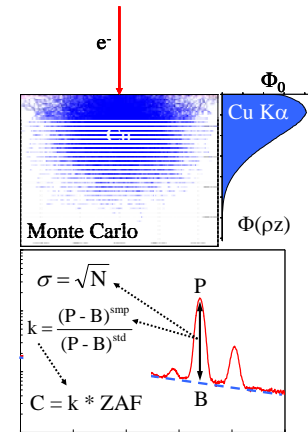
Microanalysis Software

- ◆ Casino – Monte Carlo simulation program for electron scattering and x-ray $\Phi(\rho z)$ distributions.
<http://www.gel.usherbrooke.ca/casino>
- ◆ CalcZAF– A collection of ZAF and $\Phi(\rho z)$ algorithms used to convert relative x-ray intensities to concentration. Based on CITZAF (Armstrong).
<http://epmlab.uoregon.edu/calczaf.htm>
- ◆ TRYZAF – Version of CITZAF used to compare correction algorithms and data sets for given elements.
<http://www.cstl.nist.gov/div837/Division/outputs/software.htm>
- ◆ GMR – $\Phi(\rho z)$ thin-film correction program, includes continuum fluorescence correction, used to calculate K or C for multilayer thin films on substrates.

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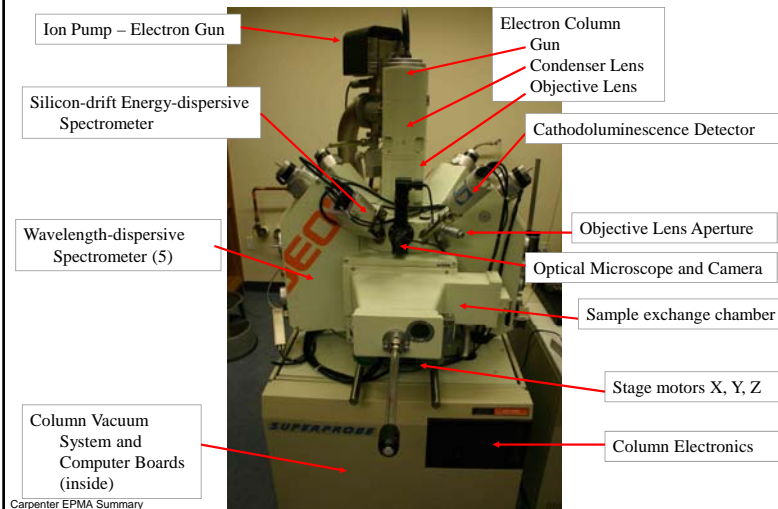
EPMA Schematic Summary

- ◆ Sample: polished, conductive
- ◆ Focused electron beam ~ 5 nm
- ◆ Electron scattering (BSE)
- ◆ Characteristic x-ray excitation ~ 1 μm analytical volume
- ◆ Generated vs. emitted x-rays
- ◆ x-ray absorption
- ◆ Comparison method k-ratio
- ◆ Correction for matrix effects: ZAF
A X-ray absorption
Z electron scattering
F X-ray fluorescence



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Electron Microprobe at Washington University



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Summary of EPMA Procedure

- ◆ Sample prep and carbon coating
- ◆ Sample imaging and energy-dispersive analysis performed to generate base map and element list.
- ◆ Microprobe stabilized, software setup, elements and standards selected.
- ◆ Wavelength spectrometers peaked on standards and PHA settings confirmed.
- ◆ Standard count rates acquired on primary and secondary standards. Evaluate accuracy with secondary standards.
- ◆ Locations on samples identified using backscattered-electron imaging and energy-dispersive spectrometer, points digitized.
- ◆ X-ray measurements acquired on samples.
- ◆ ZAF (i.e., $\Phi(\rho z)$) corrections made on samples using assigned standards.
- ◆ Analyses inspected: total, stoichiometry, etc.
- ◆ Run finished, data dumped to Excel.

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What Materials Are Analyzed by EPMA? Sample Requirements

- ◆ Samples must be:
 - Solid (EPMA does not include VP conditions)
 - Polished flat ($\sim 0.25\mu\text{m}$)
 - Conductive or made so by evaporation of ~ 25 nm carbon
 - Stable in vacuum and under electron beam bombardment
- ◆ Materials historically analyzed:
 - Elements Be-U
 - Inorganics, metals/alloys, ceramics, minerals, glasses
 - Biological, pharmaceutical, organic, etc. if fixated/stabilized
- ◆ Bulk materials (homogeneous within scattering volume)
 - Also particles, thin films/multilayers, rough surfaces, tilted
 - All deviations from flat and normal to beam must be treated with specific modifications to correction algorithms

Carpenter EPMA Summary

Sample Preparation Issues Polishing and Coating

- ◆ EPMA requires flat polished sample, usually mechanical polishing to ~ 0.25 μm final polish.
- ◆ Biological materials other preparations methods, surface topography, results in differential x-ray absorption
- ◆ SEM imaging use Au or metal sputter coating or heavy metal fixation for high SEI emission.
 - Not preferred for microanalysis due to thickness variation
 - Vacuum $\sim 10^{-3}$ Torr, thermal damage of biological matls.
- ◆ EPMA microanalysis use carbon high vacuum evaporation
 - Desire equal coat on samples and standards
 - Minimum x-ray absorption most x-rays (except Nitrogen)
 - Electron retardation affects overvoltage of energetic x-rays
 - Vacuum $\sim 10^{-6}$ Torr, no thermal damage except radiant

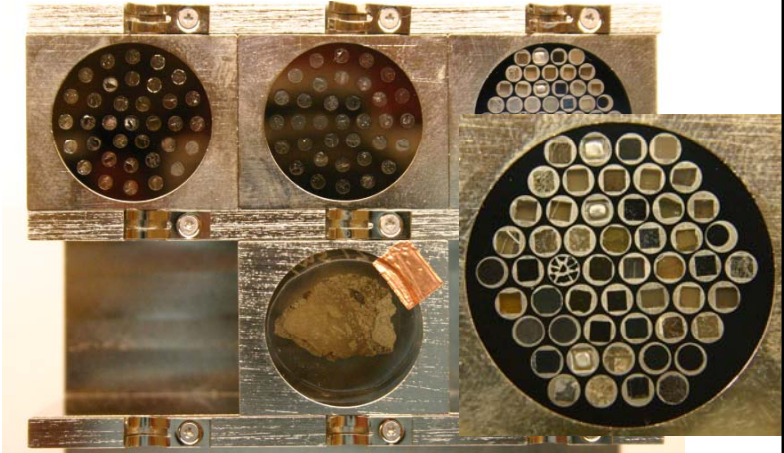
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High Vacuum Carbon Evaporator



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Microprobe Sample Holder and Standard Mount



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EPMA Summary Matrix Effects and Quantitative EPMA

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Summary of EPMA

- ◆ “Electron-probe microanalysis”, EPMA, generically means:
 - Quantitative X-ray microanalysis using a focused electron beam
 - Wavelength-dispersive spectrometers (WDS)
 - Optical microscope for sample positioning
 - Energy-dispersive spectrometer (EDS), ...*not standardless*
 - Stabilized electron column (gun shift, tilt, probe current stabilization)
- ◆ Characteristic x-rays measured on flat, polished (bulk) material at known x-ray takeoff angle to spectrometer. Elements Be-U at ~0.01-100 wt%
- ◆ Peak intensities background and deadtime corrected, precision $\sigma = \text{sqrt}(N)$ for N number of counts.
- ◆ Comparative technique: Ratio of x-ray intensity on sample is made to same on standard (“k-ratio”) using peak intensities with background subtracted:

$$k = (P-B)_{\text{sample}} / (P-B)_{\text{standard}}$$
- ◆ Measured x-ray intensities are converted to concentration units via:

$$C_{\text{sample}} = C_{\text{std}} K * Z A F, \quad K = (P-B)_{\text{sample}} / (P-B)_{\text{standard}}$$
 The ZAF correction is iterative since it is a function of composition.
- ◆ Issues: sample volume, standards, background determination, secondary fluorescence, accuracy.

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EPMA Summary: ZAF and X-ray Correction

- ◆ The absorption factor A corrects for x-ray absorption.

Beers Law absorption:

$$I = I_0 e^{(-\mu \rho \rho s)}$$
 I emitted X-ray intensity, I_0 generated X-ray intensity
 Path length ρs and mass absorption coefficient μ/ρ

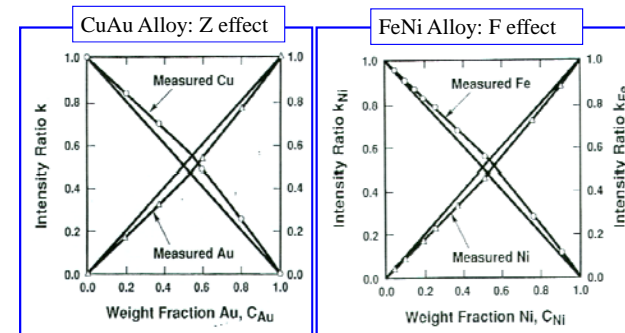
- ◆ Absorption increases with MAC and path length (...TOA, keV)
 MACs Al absorber: Cu L α 1452, Al K α 402, Cu K α 47
 MACs Cu absorber: Cu L α 1590, Al K α 4561, Cu K α 50
- ◆ Absorption generally decreases with increasing X-ray energy, except at absorption edge of matrix element(s).
- ◆ Absorption occurs within all materials, sample and standard.
- ◆ This is the largest correction ~10% for typical materials.
 AlCu alloy: Al K α absorbed by Cu
 NiFe alloy: Ni K α absorbed by Fe
 Fe K α fluoresced by Ni K α
- ◆ We measure **Emitted** intensity (I) but **Generated** (I_0) intensity is required for quantitative correction.
- ◆ Need to know generated intensity, unseen, unmeasured.
 ... Fundamental reliance on algorithms and data sets

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EPMA Summary: ZAF and X-ray Correction

- ◆ “Matrix effects”: All elements influence electron scattering/retardation, x-ray generation, absorption, and fluorescence of the element of interest, i:

$$C_i = k_i * ZAF \text{ where ZAF is effect of } C_j \text{ elements on } C_i$$



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EPMA Summary: ZAF and X-ray Correction

- Matrix effects arise because each element present in the analytical volume influences x-ray generation and propagation of every other element:
 - C_i is a function of all C_j elements
 - $C_i = k_i * ZAF$ where ZAF is effect of C_j elements on C_i
- Z: atomic number factor Z corrects for differences in (S) stopping power, i.e., electron retardation (R) electron backscattering ...between the sample and standard.
 - Example: Al K α in Al₃Cu₉₇ with avg. Z = 28.8 vs. pure element standards: Al Z = 13, Cu Z = 29
 - Expect large Z correction when matls. differ in atomic number
- A: absorption factor A corrects for x-ray absorption within both the sample and standard. This is the largest correction parameter, and is critically dependent on accelerating voltage
 - Examples: Ni K α absorbed by Fe, Al K α abs by Cu

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EPMA Summary: ZAF and X-ray Correction

- Z: atomic number factor Z corrects for differential (S) stopping power and (R) backscattering between the sample and standard.
 - Example: Al K α in Al₃Cu₉₇ with avg. Z = 28.8 vs. pure element standards: Al Z = 13, Cu Z = 29
 - Large Z correction when large difference in Z
- Note that R/S compensates so Z factor is not typically large
- ZAF factors typically presented as multiplicative factors

CalcZAF Results for Al₃Cu₉₇ Sample (also shows large A correction for Al K α) calculated at 15 keV and 40 degree takeoff angle:

ELEMENT	A	F	Z	ZAF	S	R	F(x)u	Ec	Eo/Ec
Cu ka	1.0000	1.0000	1.0024	1.0024	1.0032	.9992	.9899	8.9790	1.6706
Al ka	2.5365	1.0000	.8904	2.2586	.7872	1.1311	.3498	1.5600	9.6154

ELEMENT	K-RAW	K-VALUE	ELEMWT%	OXIDWT%	ATOMIC%	FORMULA	KILOVOL
Cu ka	.00000	.98471	98.704	-----	97.000	.970	15.00
Al ka	.00000	.00574	1.296	-----	3.000	.030	15.00
TOTAL:			100.000	-----	100.000	1.000	

C = k * ZAF, for Al: C = 0.00574 * 2.2586 = 0.01296 * 100% = 1.296 wt%

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EPMA Summary: ZAF and X-ray Correction

- NiFe Alloy, example of (A) absorption and (F) characteristic fluorescence
 - Ni K α absorbed by Fe, note MAC value 362
 - Fe K α fluoresced by Ni K α , note F value 0.78

Current Mass Absorption Coefficients From Henke (LBL, 1985) < 10KeV / CITZMU > 10KeV

Z-LINE	X-RAY	Z-ABSOR	MAC
Ni	ka	Ni	57.8
Ni	ka	Fe	362.3
Fe	ka	Ni	89.3
Fe	ka	Fe	68.3

ELEMENT	ABSCOR	FLUCOR	ZEDCOR	ZAFCOR	STP-POW	BKS-COR	F(x)u	Ec	Eo/Ec
Ni ka	1.0060	1.0000	.9987	1.0047	.9994	.9993	.9824	8.3330	1.8001
Fe ka	1.0039	.7768	1.0135	.7903	1.0056	1.0079	.9808	7.1120	2.1091

ELEMENT	K-RAW	K-VALUE	ELEMWT%	OXIDWT%	ATOMIC%	FORMULA	KILOVOL
Ni ka	.00000	.90021	90.441	-----	90.000	.900	15.00
Fe ka	.00000	.12095	9.559	-----	10.000	.100	15.00
TOTAL:			100.000	-----	100.000	1.000	

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EPMA Summary
Electron-Specimen Interactions
and
Generation and Emission of X-rays

Resolution in EPMA

- ◆ There are two aspects to “resolution” in EPMA:
- ◆ Electron scattering and the emission of secondary and backscattered electrons govern the SEI and BSE image signals
- ◆ The scattering volume is a function of average atomic number Z_{avg}
- ◆ The x-ray generation “volume” is a contoured volume depending on the excitation energy E_c for a given x-ray line
- ◆ Compare Cu Ka and Al Ka in an AlCu alloy

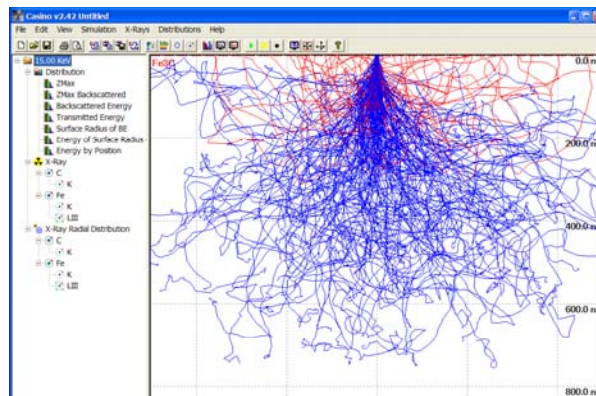
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Monte Carlo Modeling

- ◆ Monte Carlo simulations model electron-specimen interactions and x-ray production in the specimen.
- ◆ Electron scattering, slowing down, backscattering, x-ray production, etc. are all modeled using random number generation and random electron scattering directions.
- ◆ Good agreement is observed between Monte Carlo data and experimental measurements.
- ◆ This agreement is evidence of the accuracy of quantitative analysis algorithms.

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Casino Monte Carlo Program



Simulation Fe_3C , 15 keV, 5000 trajectories, max depth ~750nm
 Red = backscattered e-, blue e- within Fe_3C .
 Scattering angle increases as e- energy decreases (curly trajectory)

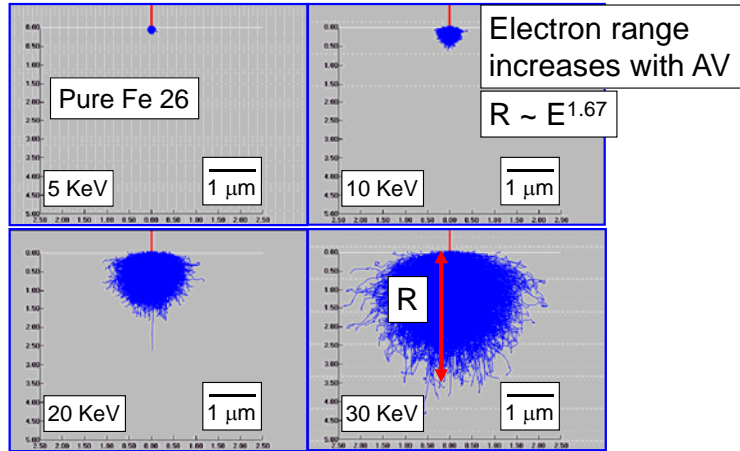
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Monte Carlo Software - Casino, DTSA II

- ◆ Casino:
 - Scattering volume
 - $\Phi(\rho z)$ curves for generated vs. emitted X-rays (K vs. L-lines)
 - Electron data, e.g. BSE energy distribution for Al vs. Au targets
 - Bulk vs. Layer geometry or grain boundary
- ◆ DTSA-II
 - EDS spectrum simulation: Experimental vs. Calc Spectra
 - Monte Carlo vs. Algorithmic X-ray simulation
 - Powerful scripting capabilities: Comp, Geometry, Structure

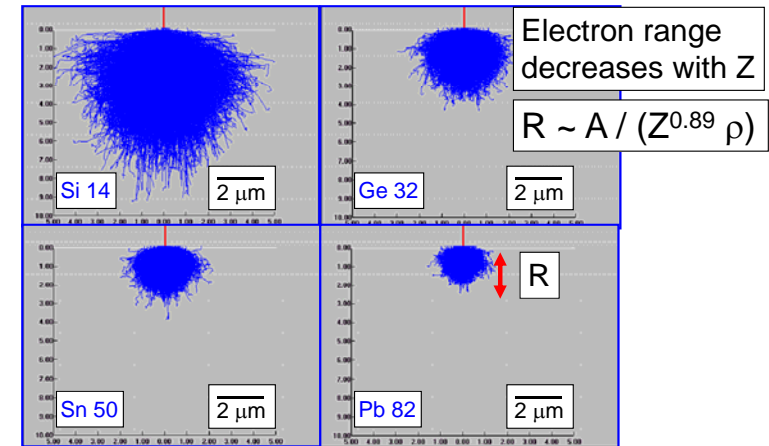
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Monte Carlo Simulation Effect of Varying Accelerating Voltage



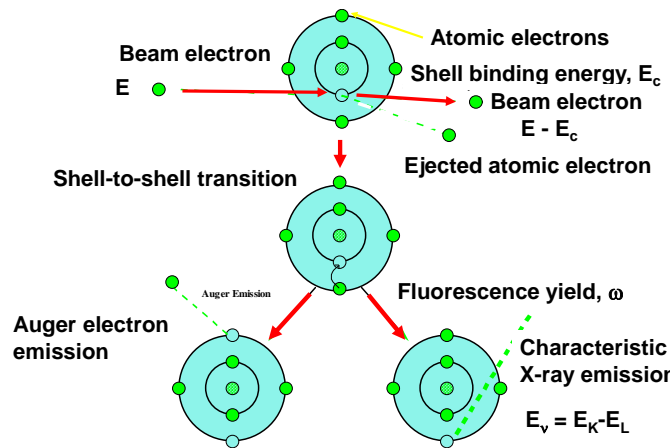
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Monte Carlo Simulation Effect of Varying Atomic Number Z @ 25 keV



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The Physics Underlying the Spectrometries

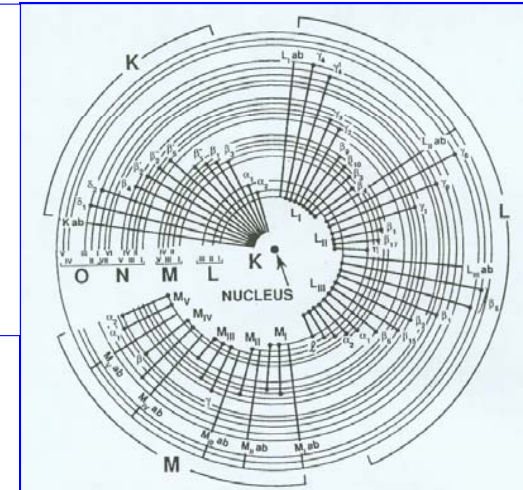


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Families of X-rays

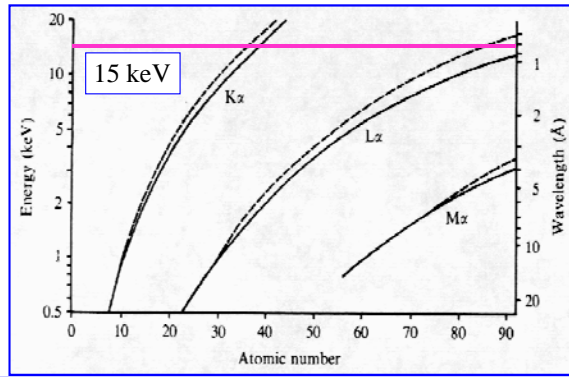
X-ray Spectra

Many ionization vacancy-filling possibilities from complex atoms lead to creation of families of X-ray lines, e.g.,
 $K\alpha - K\beta$
 $L\alpha - L\beta - L\gamma - L\eta - L\iota$
 $M\alpha - M\beta - M\gamma - M\zeta - M\eta - M\iota$



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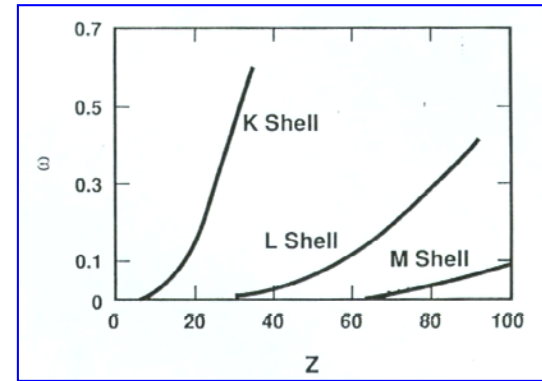
Characteristic X-ray Energy vs. Z



X-ray peak energy solid lines
 Excitation energy dashed lines
 At 15 keV at least one line can be excited for all Z

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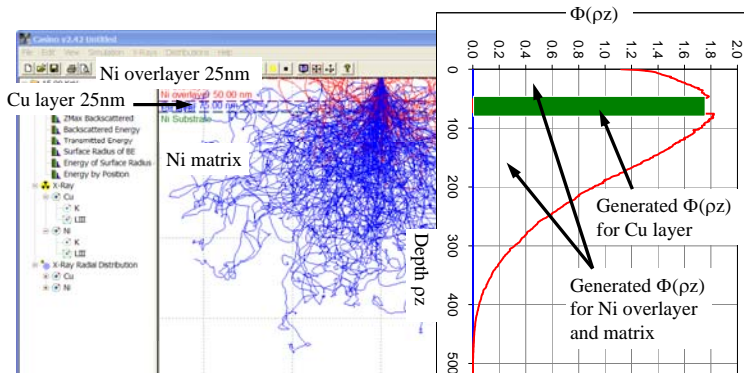
Fluorescence Yield ω vs. Z



ω = fraction of ionizations resulting in characteristic x-ray emission relative to total number of ionizations, and $1-\omega$ is the fraction of auger electrons.

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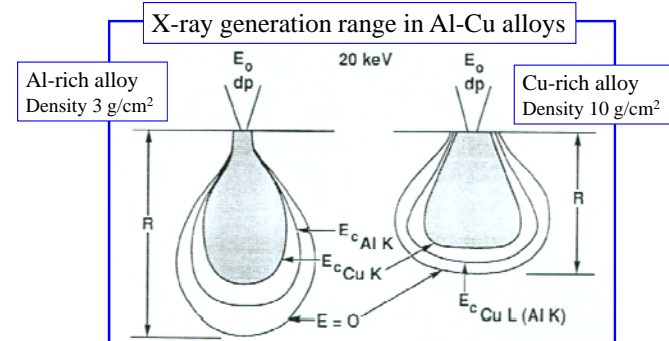
Casino MC Simulation of $\Phi(\rho z)$ Experiment



The $\Phi(\rho z)$ distribution for Cu $K\alpha$ is obtained from the emitted intensity of a layer of Cu sandwiched in a matrix with similar Z (Ni) so that scattering properties are similar. Measurements are made for progressively buried layers. This intensity is scaled relative to that from an isolated Cu foil to yield the $\Phi(\rho z)$ curve. The curves are then generalized to model x-ray behavior for elements in general.

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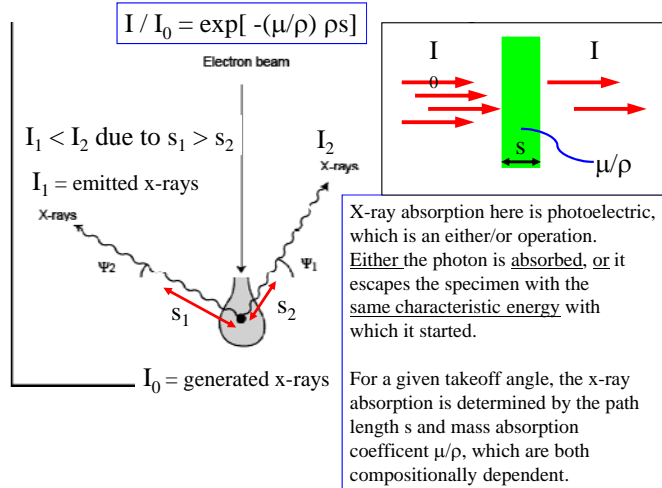
Range of Characteristic X-ray Production



The depth distribution of primary x-ray generation depends on Z_{avg} and the relative excitation energies E_c of the elements in the sample:
 Al $K\alpha$ $E_c = 1.559$ keV generated in larger volume $E > 1.559$ keV
 Cu $K\alpha$ $E_c = 8.98$ keV, generated in small volume $E > 8.98$ keV

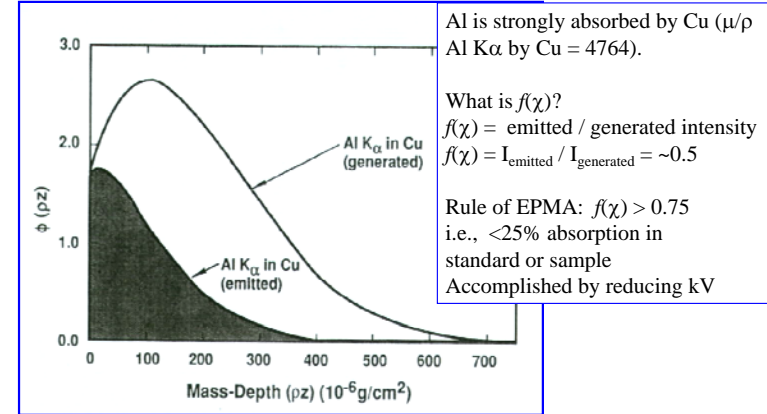
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X-ray absorption depends on path length and μ/ρ



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Effect of X-ray Absorption: Trace Al in Cu @ 20 keV



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EPMA Summary: ZAF and X-ray Correction

- ◆ Z: atomic number factor Z corrects for differential (S) stopping power and (R) backscattering between the sample and standard.
 Example: Al $K\alpha$ in Al_3Cu_{97} with avg. $Z = 28.8$ vs. pure element standards: Al $Z = 13$, Cu $Z = 29$
 Large Z correction when large difference in Z
- ◆ Note that R/S compensates so Z factor is not typically large
- ◆ ZAF factors typically presented as multiplicative factors

CalcZAF Results for Al_3Cu_{97} Sample (also shows large A correction for Al $K\alpha$)
 calculated at 15 keV and 40 degree takeoff angle:

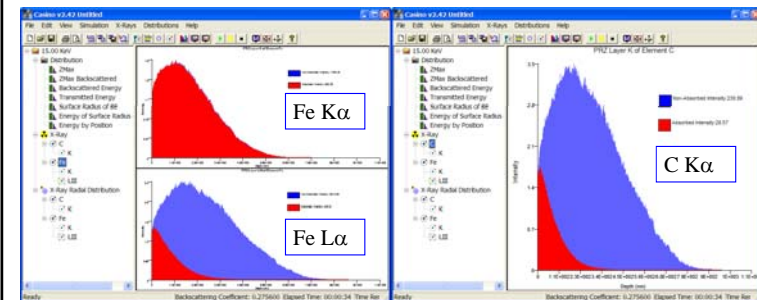
ELEMENT	A	F	Z	ZAF	S	R	F(x)u	Ec	Eo/Ec
Cu $K\alpha$	1.0000	1.0000	1.0024	1.0024	1.0032	.9992	.9899	8.9790	1.6706
Al $K\alpha$	2.5365	1.0000	.8904	2.2586	.7872	1.1311	.3498	1.5600	9.6154

ELEMENT	K-RAW	K-VALUE	ELEMWT%	OXIDWT%	ATOMIC%	FORMULA	KILOVOL
Cu $K\alpha$.00000	.98471	98.704	----	97.000	.970	15.00
Al $K\alpha$.00000	.00574	1.296	----	3.000	.030	15.00
TOTAL:			100.000	----	100.000		

$C = k * ZAF$, for Al: $C = 0.00574 * 2.2586 = 0.01296 * 100\% = 1.296 \text{ wt}\%$

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Casino $\Phi(\rho z)$ Curves Calculated for Fe_3C



Casino Monte Carlo calculated $\Phi(\rho z)$ curves
 Generated x-ray distribution in blue, Emitted x-ray distribution in red
 These intensities can be used to calculate k-ratios and compared with k-ratios from experimental and $\Phi(\rho z)$ algorithms

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Backscattered Electron Imaging and X-ray Mapping

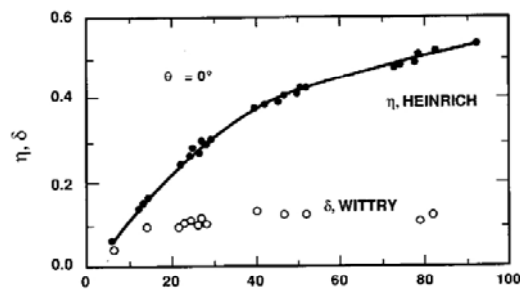
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Contrast Mechanism

- ◆ Changes in the signal(s) (“Contrast”) collected at different points in the image convey information about specimen characteristics.
- ◆ Whenever a signal changes in a predictable way as a function of specimen properties, we have the basis for a contrast mechanism.
- ◆ Example: η vs. Z is monotonic and predictable. This forms the basis for atomic number (compositional) contrast.

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Backscattered Electron Dependence on Z Atomic Number



Backscattered electrons: primary beam electrons scattered at high angles by the nucleus of atoms in the sample. The fraction of BSE is eta and is a function of atomic number Z for pure elements and average Z for compounds. The BSE signal is used for mineral ID.

Secondary electrons are emitted from the valence shell of atoms and are not sensitive to the Z of the material, so are not useful for identifying phases

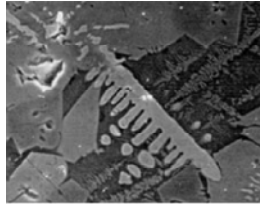
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Mean Atomic Number of Minerals

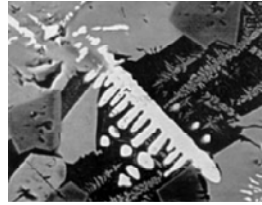
- ◆ Mean Z can be easily calculated (CalcZAF, for example)
- ◆ Typical range is $\sim 10 - 20$. Atomic number correction is limited if appropriate standards are used.
- ◆ Silicates
Forsterite 10.6 Albite 10.7 Pyrope 10.7
Fayalite 18.7 Anorthite 11.9 Almandine 15.6
- ◆ Oxides
Periclase 10.4 Spinel 10.6 Quartz 10.8
Ilmenite 19.0 Ulvospinel 20.0 Magnetite 21.0
- ◆ Sulfides
Pyrite 20.7 Chalcocite 26.4 Galena 73.2
- ◆ Carbonates
Dolomite 8.9 Calcite 12.4 Siderite 16.5 Cerussite 65.3
- ◆ Misc
Apatite 14.1 Allanite 22.0 Zircon 24.8 Monazite 38.7
- ◆ See SJB Reed Electron Microprobe Analysis and Scanning Electron Microscopy in Geology ISBN 0521483506 (paperback)

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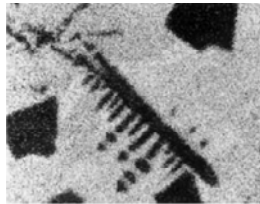
Image Signals from Metallographic Sample



Secondary electron



Backscattered electron



Silicon x-ray



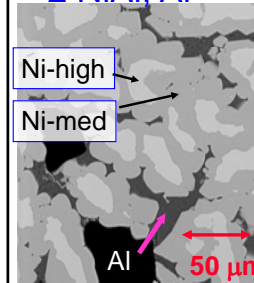
Aluminum x-ray

20 kV 1100x, JEOL

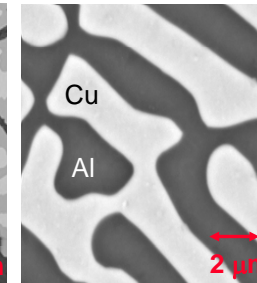
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Atomic Number Contrast

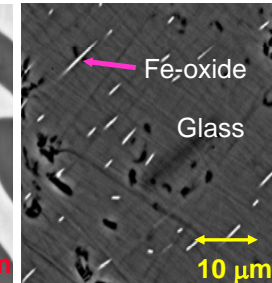
Raney Ni-Al
2 NiAl, Al



Al-Cu eutectic



Obsidian



Backscattered-electrons provide a rapid and simple means to image differences in chemistry in the sample, via the dependence on average atomic number.

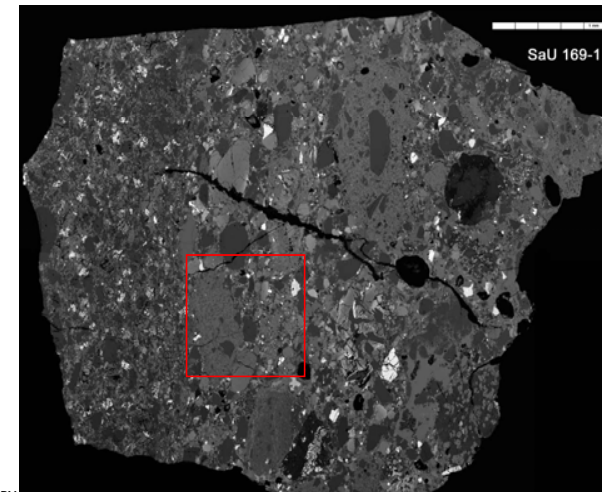
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Compositional Mapping

- ◆ Normal x-ray mapping is performed by setting up an energy region of interest (ROI) in the EDS spectrum. All x-rays falling in that ROI are assigned to the digital map for that element.
- ◆ Drawback is each map has only information for that element and can not be used to perform quantitative analysis. X-rays from other elements also may be counted in the ROI – peak overlaps, continuum x-rays, etc.
- ◆ Sophisticated mathematical approaches have been used in recent years to process x-ray map data. These involve correlation between elements, but also correlation between groups of elements.
- ◆ Examples of these approaches are Concentration Histogram Imaging and Principle Component Analysis.
- ◆ Spectrum Imaging is a mapping technique where the entire EDS spectrum is saved at each pixel location and is available for processing after the run.

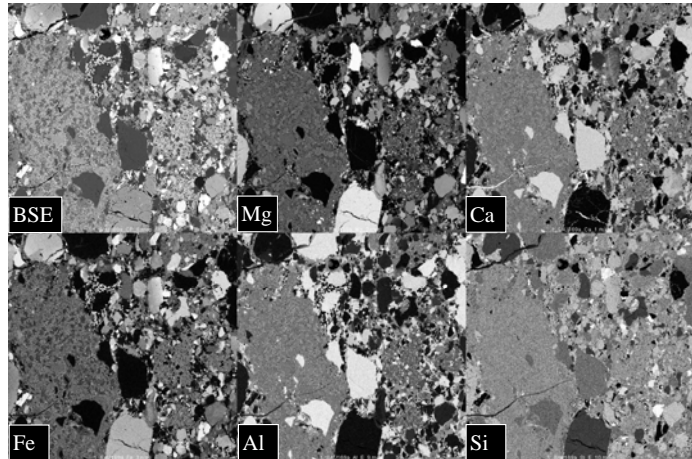
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Lunar Meteorite SaU 169 BSE Mosaic



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Lunar Meteorite SaU 169 SDD X-ray Maps



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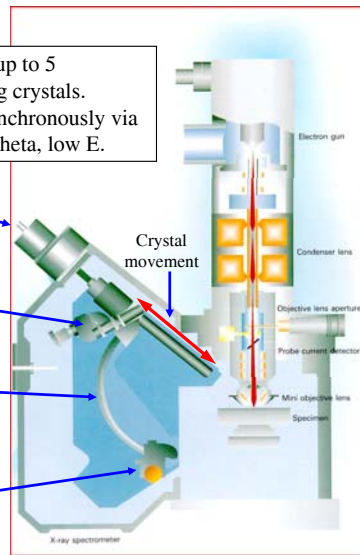
EPMA Summary Wavelength-Dispersive Spectrometry

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WDS Anatomy

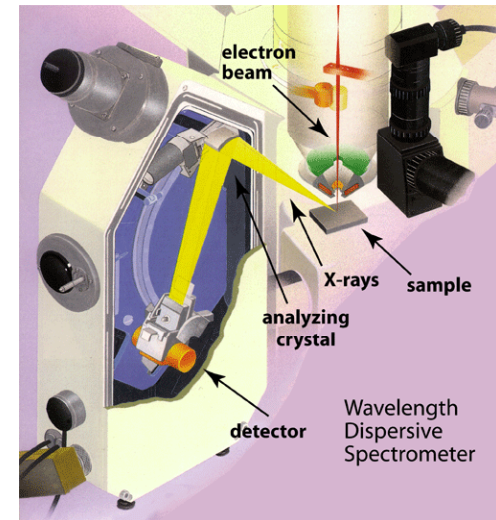
An electron microprobe typically has up to 5 spectrometers each with 1-4 diffracting crystals. The crystal and detector are moved synchronously via the drive motor. Position here is high theta, low E.

- Spectrometer drive motor w/ vacuum feedthrough
- Crystals w/ flipping motor
- Mechanical alignment track
- Proportional X-ray counter



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WDS Schematic JEOL Electron Microprobe

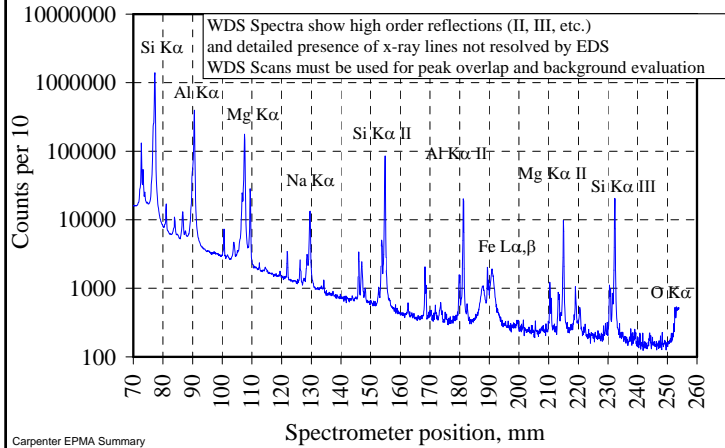


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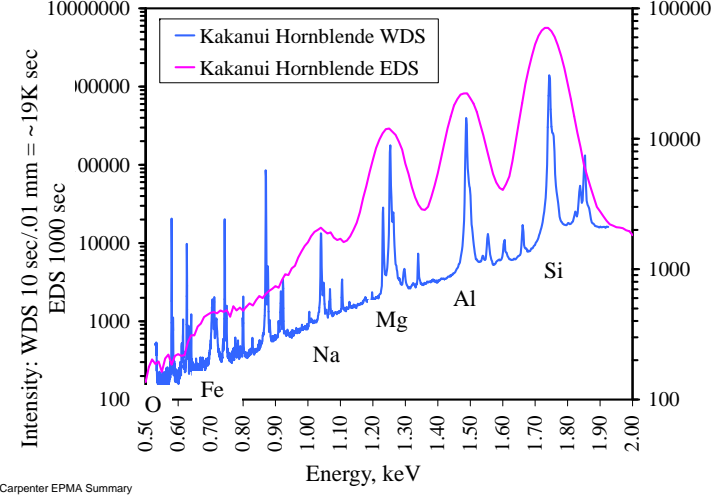
WDS Wavelength Scan

TAP Spec 2 25 KV 300 nA 70 - 266 mm 10 sec per point 0.1 / .09 mm step

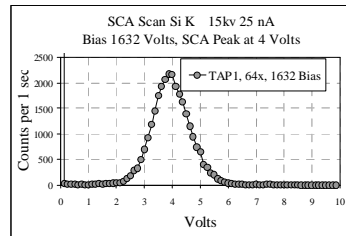
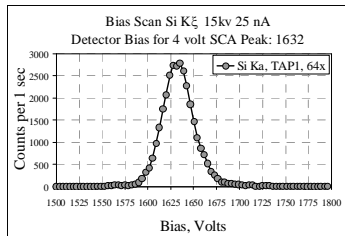
WDS Scan TAP Kakanui Hornblende



WDS and EDS Comparison: Kakanui Hornblende



Pulse-height Analysis: WDS



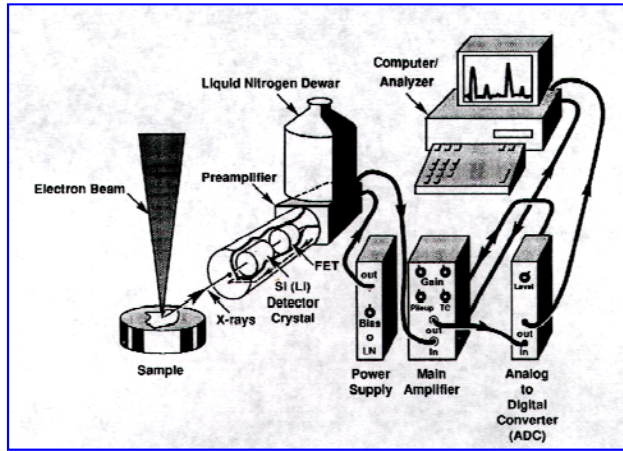
Detector bias scan for Si Kα using 3.9 volt baseline and 0.2 volt window. The bias scan is used to set the detector bias necessary to achieve a 4 volt pulse on an SCA scan (from 0-10 Volts). Bias scans are typically performed in the 1500-1800 Volt range, which is the plateau region of the x-ray detector. This is for uniform application of detector gain and consistent detector deadtime behavior.

SCA (single channel analyzer) scan over detector voltages produced by entry of Si Kα photons into the gas-flow counter when the WDS is set to the Si Kα peak position. The detector bias is set to the value determined in the bias scan, which ensures that the SCA pulse is at the nominal value (here 4 Volts). An SCA scan is typically performed over the range 0-10 Volts. This uniform application of detector gain enables the use of uniform baseline and window values, and also minimizes the effect of gain variations on deadtime. Note the baseline noise below 0.25 Volts. At high count rates the pulse will shift to lower values and will eventually merge into the baseline due to inability of the counting electronics to resolve the pulse energy. For this reason it is important to monitor the bias and SCA voltage values used for an element.

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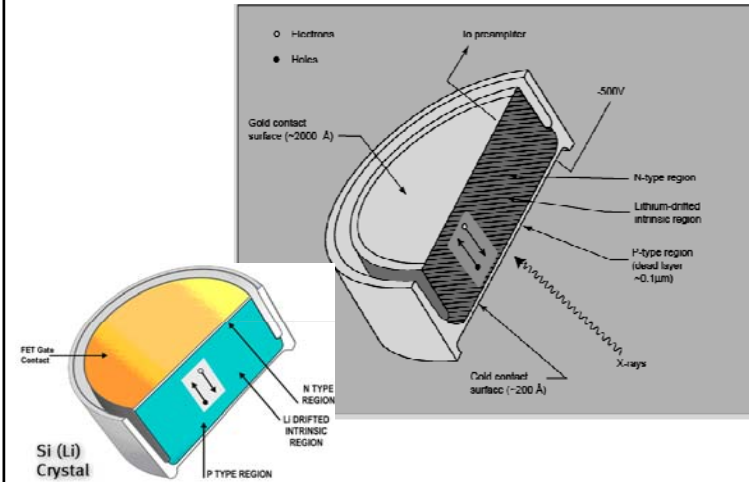
EPMA Summary
Energy-Dispersive Spectrometry
Silicon Drift EDS Detectors
Cathodoluminescence

EDS Analyzer System



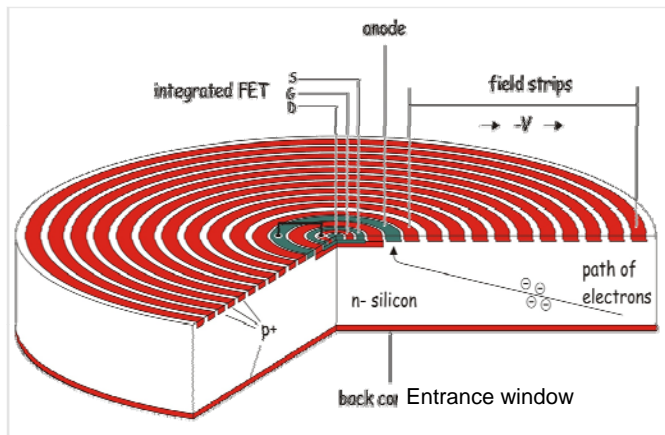
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Conventional SiLi EDS Detector



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Silicon Drift Detector



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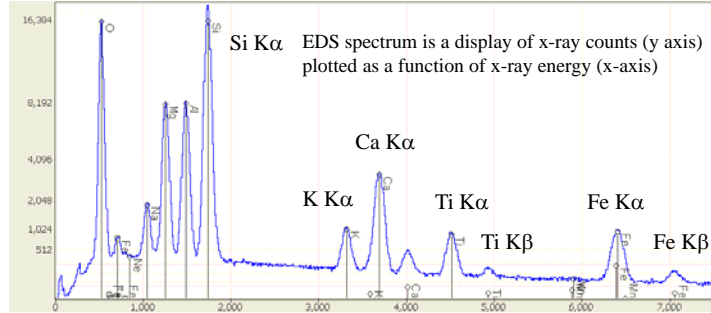
Washington University Earth and Planetary Sciences JEOL JXA-8200



JEOL e2v Silicon Drift Detector
 130 eV resolution
 3 time constants T3 T2 T1
 Stage and beam mapping
 Quantitative EDS analysis LLSQ

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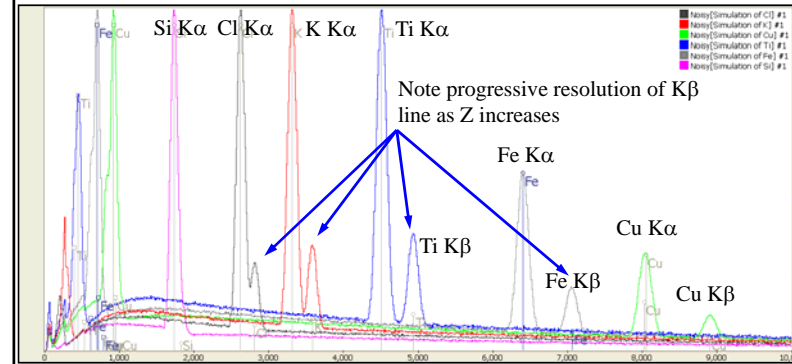
Kakanui Hornblende EDS X-ray Spectrum Generated Using DTSA-II



The EDS spectrum has **characteristic** x-ray peaks produced by specific transitions from the atom's excited state that are specific to each element. These characteristic peaks are superimposed on **continuum** ("Bremsstrahlung") produced by decelerating electrons. The EDS spectrum is used for routine identification of phases and for analysis. Note increasing energy resolution with increasing x-ray energy (Si vs. Fe) Peak overlaps (Na-Si K α peaks, and n-1 element K β peak on n element K α (K, Ca)

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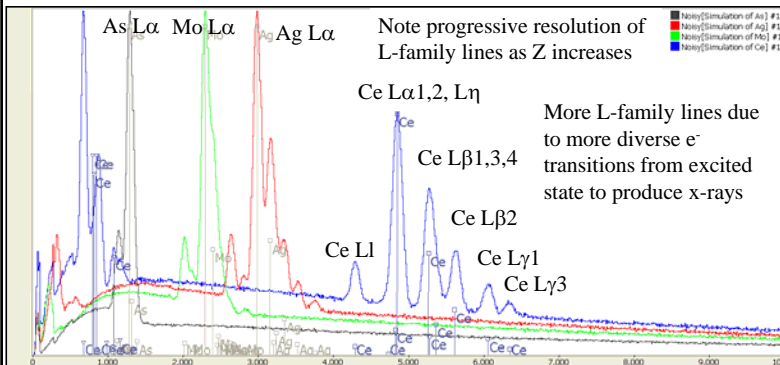
K-Family X-ray Spectra (EDS) Generated Spectra Using DTSA-II, $E_0 = 15$ keV



Also note appearance of L-family lines as Z increases

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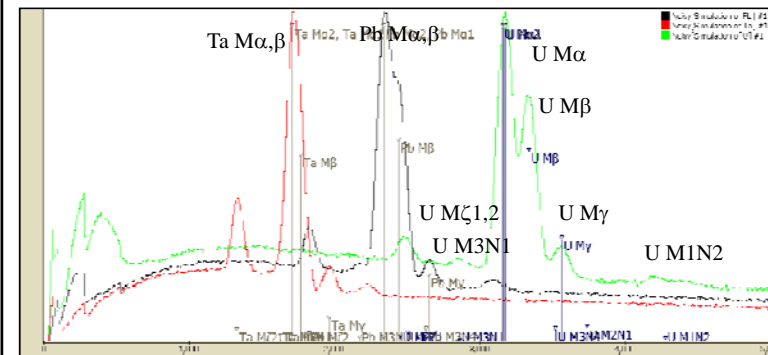
L-Family X-ray Spectra (EDS) Generated Using DTSA-II, $E_0 = 15$ keV



Also note appearance of M-family lines as Z increases

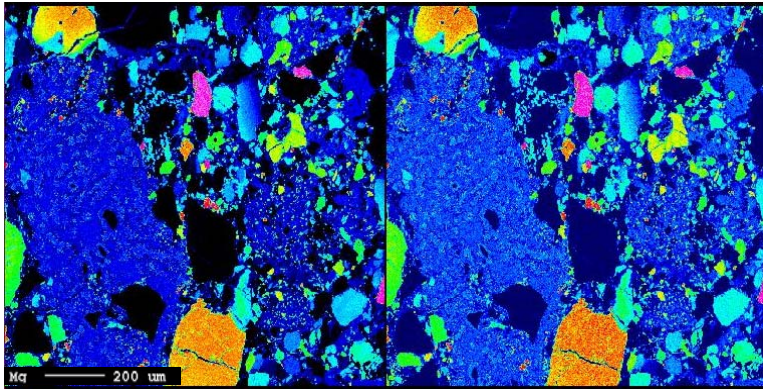
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M-Family X-ray Spectra (EDS) Generated Using DTSA-II, $E_0 = 15$ keV



Carpenter EPMA Summary

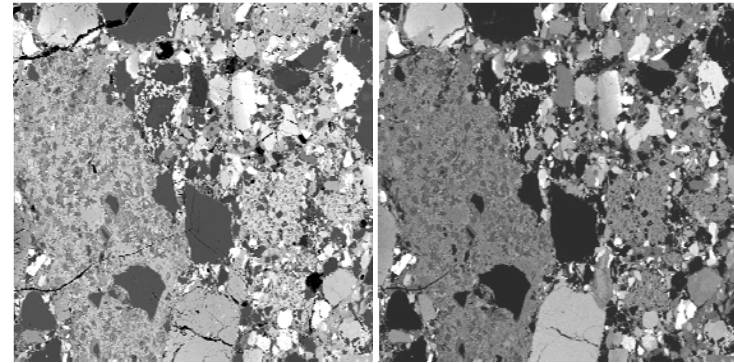
JEOL 8200 Stage Maps: Lunar Meteorite SAU169
WDS vs. SDD Mg @ 15KV, 120nA, 25ms dwell



Mg WDS 1061 max counts, Mg SDD 527 max counts
1024x1024 stage map, 8 hours

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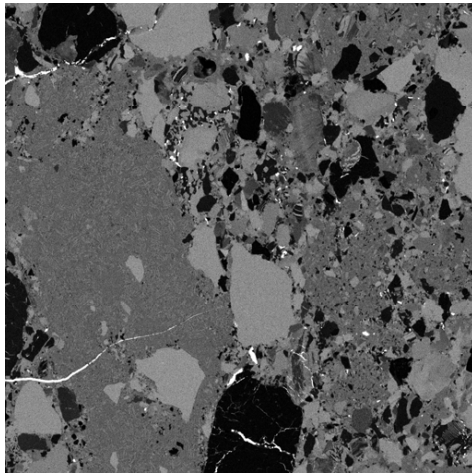
Lunar Meteorite SAU169
Stage map 1024x1024, 25 ms, 8 hr run



Backscattered electron vs. Fe SDD maps

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Lunar Meteorite SAU169
Stage map 1024x1024, 25 ms, 8 hr run: Ca K α
SDD



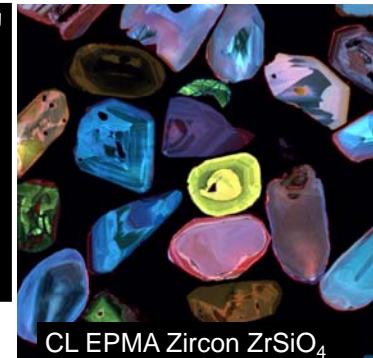
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Cathodoluminescence Advances

Luminoscope CL image of typical single Crystal
Diamond, South Africa



HFW ~3 mm (courtesy of S. Shirey)



CL EPMA Zircon ZrSiO₄

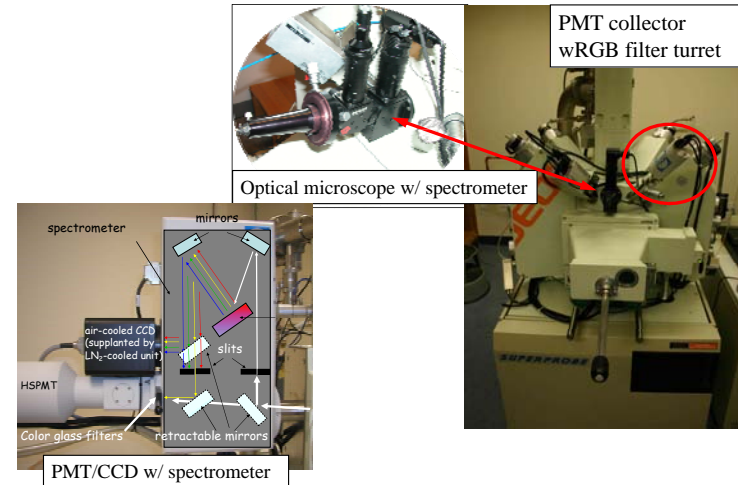
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Cathodoluminescence

- ◆ CL is the emission of photons in the visible light region resulting from electron bombardment; EPMA with focused beam is excellent for CL
- ◆ The spectroscopy and interpretation of CL are complex
- ◆ CL is produced by defects (semiconductors) and/or trace elements (minerals)
- ◆ Activation can be produced by trace elements at ppb-ppm concentrations, well below nominal EPMA sensitivity (bse, x-ray)
- ◆ Minerals: activation by Mn, REE, etc., quenched at high concentration and by Fe
- ◆ CL collected by photomultiplier, parabolic mirror, fiber optic/grating systems
- ◆ Total CL signal vs. spectral and hyperspectral data collection
- ◆ Quantification requires spectrum peak deconvolution

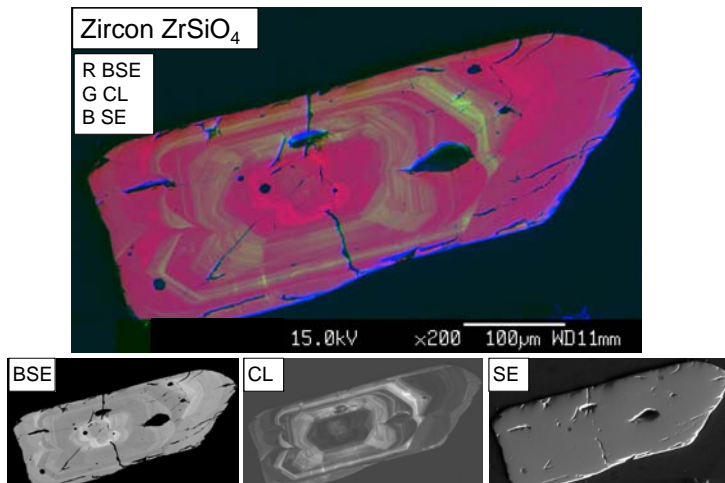
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Types of EPMA CL systems



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Combined CL, BSE, SE Images in Zircon



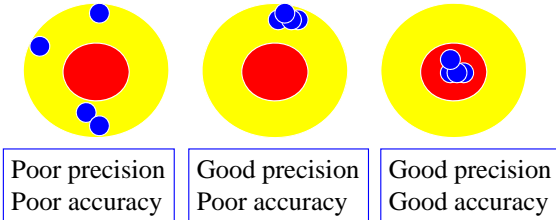
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EPMA Summary
Precision and Accuracy
in
Microanalysis

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Precision vs. Accuracy

- ◆ Precision is the reproducibility of replicate measurements. High precision means reproducible. Target practice: grouping of impact points.
- ◆ Accuracy is the correctness of the measurement. Target practice: proximity to bullseye.



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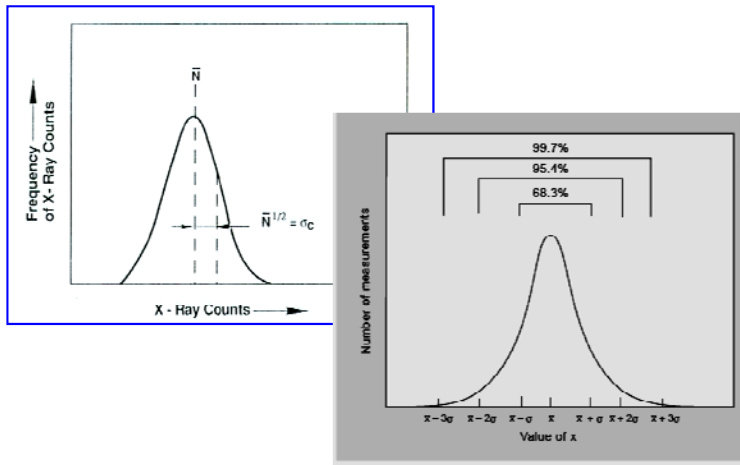
What is the Precision of Analysis?

- ◆ Precision is the reproducibility of a measurement, which accuracy is the degree of correctness of a measurement.
- ◆ X-ray production is statistical, and counting experiments yield a gaussian distribution of counts for which the standard deviation σ_c is the square root of the mean of the counts N_m :

$$\sigma_c = (N_m)^{1/2}$$
- ◆ σ_c is 1-sigma (67% confidence level), multiply by 2 for 2-sigma (97%), by 3 for 3-sigma (99%).
- ◆ Without drift or other instrumental instabilities, the precision of the measurement is determined by the number of x-rays counted.
- ◆ In the case of EDS, the relatively poor P/B makes it unclear as to what is the "peak" and what is the "background". Counts are obtained from the fitting procedure. Peak channel has highest P/B and is not the whole story.

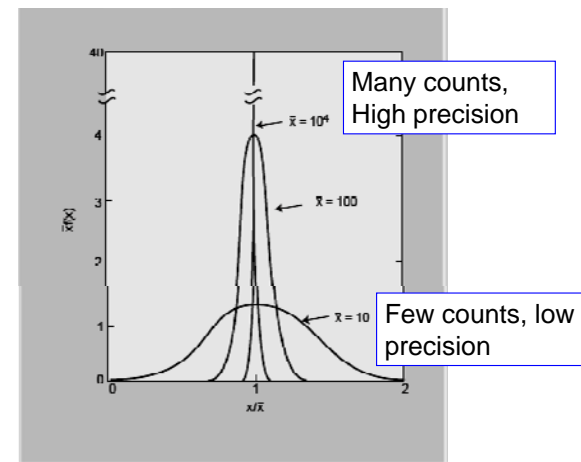
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Counting Statistics—Gaussian Distribution



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Relative Precision as a Function of Number of X-rays Counted



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Precision of X-ray Counting

Counts N_m	σ_c	$\sigma_c / N_m * 100$ (relative percent)
100	10	10 %
1000	31.62	3.16 %
10,000	100	1.0 %
100,000	316.2	0.316 %
1,000,000	1000	0.1 %

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Actual Standard Deviation

- The actual standard deviation of a set of measurements is the sum of the standard deviation due to counting statistics alone, plus the actual variation of an element in a sample, plus any other factors that cause variation (drift, for example):

$$\sigma_{\text{Total}} = \sigma_{\text{count stats}} + \sigma_{\text{actual}} + \sigma_{\text{drift}} + \dots$$

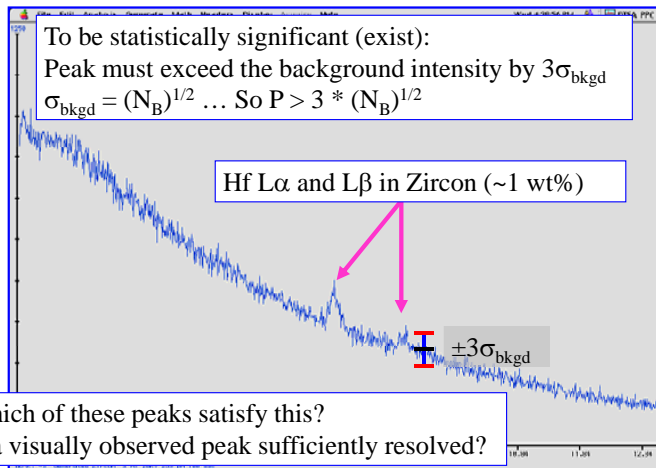
- The actual standard deviation is:

$$S_c = \left[\sum_{i=1}^n \frac{(N_i - \bar{N})^2}{(n-1)} \right]^{1/2}$$

where n is the number of measurements and N bar is the mean.

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Are These Peaks Real?



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Detection Limit

- An element is present when the peak exceeds 3 times the standard deviation of the background intensity measured at the peak position. This is a statistical approach, 3 σ is 99% of the background variation and any counts outside of that range must be a "peak".

$$N_p > 3(N_b)^{1/2}, N_p \text{ is peak, } N_b \text{ is background}$$

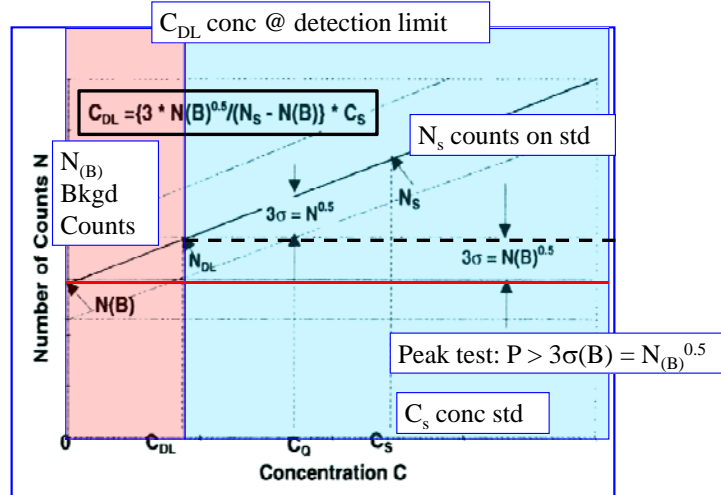
- A convenient formula for calculation of detection limit is that of Ziebold (1967):

$$C_{DL} \geq \frac{3.29a}{(ntP \cdot P/B)^{1/2}}$$

where t is the time of measurement, n is # of repetitions, P is count rate on pure element, P/B is peak-to-background of pure element, and a is the c/k factor (i.e. ZAF). P^2/B is the figure of merit.

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Detection Limit



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Detection Limit EDS vs. WDS

Table 9.17. Comparison of the Minimum Detectability Limit of Various Elements Using an EDS and WDS Detection System on the Basis of Optimized Operating Conditions*

Analysis	Element	P (cps)	B (cps)	P/B	Wet chem. (wt %)	C_{DL} (wt %)
EDS	Na $K\alpha$	32.2	11.5	2.8	3.97	0.195
	Mg $K\alpha$	111.6	17.3	6.4	7.30	0.102
	Al $K\alpha$	103.9	18.2	5.7	4.67	0.069
	Si $K\alpha$	623.5	27.3	22.8	26.69	0.072
	Ca $K\alpha$	169.5	19.9	8.5	12.03	0.085
WDS	Na $K\alpha$	549	6.6	83	3.97	0.021
	Mg $K\alpha$	2183	8.9	135	7.30	0.012
	Al $K\alpha$	2063	16.1	128	4.67	0.008
	Si $K\alpha$	13390	37.0	362	26.69	0.009
	Ca $K\alpha$	2415	8.2	295	12.03	0.009

* Geller (1977). Analysis of DI-JD-35; EDS data collected at 2000 cps for 180 s, dead-time corrected (25%) (1.75 nA probe current at 15 keV); WDS data collected for 30 s for each element; 180 s total analysis time (30 nA probe current at 15 keV).

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What is the Accuracy of Microanalysis?

- ◆ Precision is determined by the total number of x-rays counted for a given element. In general, measurements are made to better precision than accuracy warrants.
- ◆ Do not confuse precision with accuracy: replication of measurements does not tell you anything about the accuracy.
- ◆ Accuracy is dependent on all aspects of measurement:
 - Standards—how well characterized are the compositions?
 - Measurement process—systematic errors (takeoff angle?) instrument stable? Same conditions for std and smp?
 - Peak fitting procedure—calibration differences sample vs. standard, other artifacts of peak stripping?
 - X-ray correction algorithms and data sets—how accurate are the calculated results given “perfect” measurements?
- ◆ There is no global “accuracy” for EPMA measurements!

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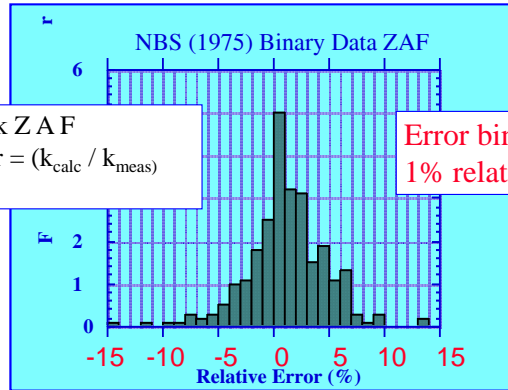
How Can Accuracy Be Determined?

- ◆ Analysis of a secondary standard as an unknown, under the same conditions as sample measurement, allows determination of accuracy.
- ◆ Accuracy should be calculated from:

$$\%Rel\ Acc = 100 * (Meas - Accepted) / (Accepted)$$
 The sign is important + is high, - is low.
- ◆ Evaluation of experimental binary alloy data sets using different x-ray correction algorithms yields k_{corr} / k_{exp} histograms. But what if these data sets were used to develop the correction procedure? Is this a test?
- ◆ Critical evaluation of experimental data sets suggests that for the best measurements the accuracy is ~1-2%, and is probably limited by precision.
- ◆ For “problem” elements the accuracy may be much worse, perhaps 5-10-20-50-100% !! Depends on concentration too.

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Error Distribution for Matrix Corrections with ZAF/Standards (Pure Elements, Binary Compounds)



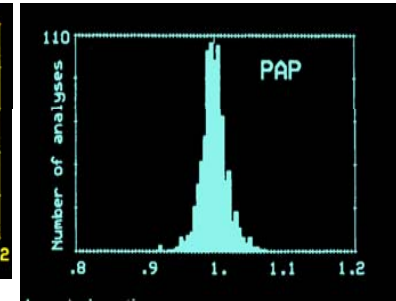
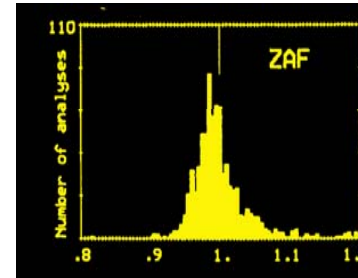
$C = C_{std} k Z A F$
 Rel Error = $(k_{calc} / k_{meas}) * 100$

Error bins are 1% relative wide

This distribution has $\sigma = 2.5\%$. 95% of analyses are expected to lie within $\pm 5\%$ relative of the correct value.

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What If the $\Phi(\rho Z)$ /standards Method Is Used Instead of ZAF/standards?



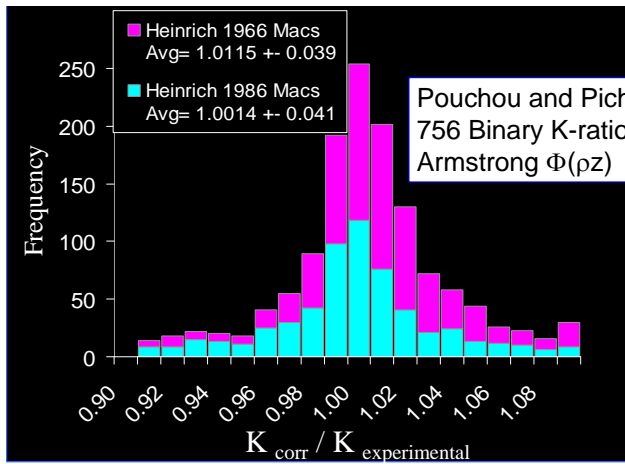
Conventional ZAF

Pouchou & Pichoir $\Phi(\rho Z)$

The relative error distribution may be narrower yet!
 The improvement depends on the particular elements.
 Light element quantitation is particularly improved.

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Heinrich 1966 vs. 1986 Mass Absorption Coefficients



Heinrich 1966 Macs
 Avg= 1.0115 +- 0.039

Heinrich 1986 Macs
 Avg= 1.0014 +- 0.041

Pouchou and Pichoir
 756 Binary K-ratio
 Armstrong $\Phi(\rho Z)$

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